

# Modelling slight compressibility for hyperelastic anisotropic materials

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## Abstract

In order to avoid the numerical difficulties in locally enforcing the incompressibility constraint using the displacement formulation of the Finite Element Method, slight compressibility is typically assumed when simulating the mechanical response of nonlinearly elastic materials. The current standard method of accounting for slight compressibility of hyperelastic materials assumes an additive decomposition of the strain-energy function into a volumetric and an isochoric part. A new proof is given to show that this is equivalent to assuming that the hydrostatic stress is a function of the the volume change only and that uniform dilatation is a possible solution to the hydrostatic stress boundary value problem, with therefore no anisotropic contribution to the mechanical response. An alternative formulation of slight compressibility is proposed, one that does not suffer from this defect. This new model generalises the standard model by including a mixed term in the volume change and isochoric response. Specific models of slight compressibility are given for isotropic, transversely isotropic and orthotropic materials.

*Keywords: nonlinearly elastic materials; slightly compressible; volumetric-isochoric split.*

# 1 Introduction

The Finite Element Method (FEM) is the preeminent numerical method when simulating the mechanical response of non-linearly elastic materials. The use of the FEM when modelling soft tissue has focussed attention on the accounting for the slight compressibility of such tissue, especially as to how it is incorporated in commercial and open-source Finite Element codes, which are used in the vast majority of simulations. Without exception, slight compressibility is modelled in these codes by assuming that the strain energy function per unit undeformed volume  $W$  can be additively decomposed into separate volumetric and isochoric components as follows:

$$W(J, \mathbf{C}^*) = f(J) + \mathcal{W}(\mathbf{C}^*), \quad (1.1)$$

where  $J$  is the determinant of the deformation gradient tensor  $\mathbf{F}$  and  $\mathbf{C}^*$  is the isochoric right Cauchy-Green strain tensor defined by  $\mathbf{C}^* = J^{-2/3}\mathbf{C}$ ,  $\mathbf{C} \equiv \mathbf{F}^T\mathbf{F}$ , noting that  $\det \mathbf{C}^* = 1$  (see, for example, Flory [1]). Typically the incompressibility constraint is to be relaxed for a known incompressible strain energy  $W_i(\mathbf{C})$ , with therefore

$$W(1, \mathbf{C}) = W_i(\mathbf{C}), \quad (1.2)$$

and so here  $\mathcal{W}(\mathbf{C}^*) = W_i(\mathbf{C}^*)$ , assuming that  $f(1) = 0$ . Ideally  $W_i(\mathbf{C})$  should be compatible with the corresponding linear theory, as it seems that a well-posed non-linear model should fully recover the linear on restriction to infinitesimal values of the independent variable. Although not necessary for what follows, it will be assumed here that this compatibility is satisfied.

The assumption (1.1) is typically given without motivation; the primary reasons for its widespread use seem to be an intuitive appeal and mathematical convenience. Another reason for its widespread use when modelling anisotropic soft tissue could be its success in modelling *isotropic* elastomers, for which the decomposition (1.1) was originally proposed;

40 it seems that the isotropic formulation of (1.1) was generalised in a natural way for  
41 anisotropic materials, without much consideration being paid as to how appropriate this  
42 is. This is the main focus here.

43 It is shown that the following are consequences of assuming (1.1) for all anisotropic  
44 materials:

- 45 • The corresponding linear theory cannot be fully recovered from the non-linear the-  
46 ory on restriction to infinitesimal strains (Federico [2]). It is axiomatic that any  
47 non-linear theory should recover the linear on restriction to infinitesimal values of  
48 the independent variable(s) (Quintanilla and Saccomandi [3]).
- 49 • Uniform dilatation is a solution to the boundary value problem of hydrostatic ten-  
50 sion. This is the solution that is simulated in FEM codes (Ní Annaidh *et al.* [4])  
51 and thus the fibres play no role in the mechanical response in this problem, which  
52 is surely physically unrealistic.
- 53 • The volumetric-isochoric split is equivalent to assuming that the trace of the Cauchy  
54 stress is a function only of the volume change (Charrier *et al.* [5]). Although the  
55 intuitive appeal of (1.1) is obvious, its equivalent formulation in terms of the stress  
56 lacks this appeal. Is it reasonable to postulate a theory of slight compressibility for  
57 anisotropic materials based on the assumption of isotropic response under hydro-  
58 static tension?

59 Attention has already been drawn to these deficiencies for specific forms of anisotropy by,  
60 for example, Ní Annaidh *et al.* [4], Vergori *et al.* [6], Gilchrist *et al.* [7] and Nolan *et al.*  
61 [8]. The novelty here is that all results are obtained in complete generality, irrespective  
62 of material symmetry. Somewhat surprisingly, it is shown that proof of the above listed  
63 results for a general elastic material is trivial in comparison to the existing proofs for  
64 specific anisotropic models, such as those obtained by Sansour [9] and Horgan and Murphy  
65 [10] for example.

66 The inability of models incorporating the volumetric-isochoric split (1.1) to capture  
67 physically realistic effects in hydrostatic tension and compression is the core element of  
68 the difficulties listed above. It might be argued that this inability is not important when  
69 simulating the mechanical response of slightly compressible materials in typical applica-  
70 tions. However, when dealing with both complex geometries and boundary conditions, it  
71 is impossible to rule out at least local states of hydrostatic tension and compression and  
72 therefore accurate accounting for the behaviour of slightly compressible materials in these  
73 experiments is essential. An alternative approach to resolving these difficulties in com-  
74 pressible elasticity could be the simulation of perfect incompressibility instead. However,  
75 this is not an option, even if the numerical difficulties that this poses can be overcome  
76 in an efficient and accurate manner, as no real material is perfectly incompressible and  
77 slight compressibility is a fundamental aspect of the physical response of materials that  
78 have classically been modelled as being perfectly incompressible, such as soft tissue.

79 An alternative formulation of slight compressibility is suggested here in order to over-  
80 come the difficulties associated with (1.1). It seems sensible to generalise this formulation  
81 of slight compressibility in order to utilise the vast computational infrastructure already  
82 developed that is based on the volumetric-isochoric split. The approach proposed here  
83 is based on truncating a Taylor series in the volume change after the second order, as  
84 initially suggested by Spencer [11]. The zero order term is a perfectly incompressible ma-  
85 terial, assumed known from standard material characterisation tests. The first and second  
86 order coefficients need to be specified. Motivated by mathematical convenience, it will be  
87 assumed that the first order term is linear in the appropriate invariants of  $\mathbf{C}^*$  and that  
88 the second order term is a positive constant. This has the intuitive appeal of a decreasing  
89 complexity in the  $\mathbf{C}^*$  terms as the order of the Taylor series in  $J$  increases. The current  
90 standard formulation of slight compressibility is a special case of this new approach, with  
91 (1.1) recovered if the linear term in the volume change is identically zero. Explicit models  
92 are proposed for modelling slightly compressible for isotropic, transversely isotropic and  
93 orthotropic materials.

## 2 Preliminaries

The constitutive law for compressible, homogeneous, non-linearly elastic materials is given by

$$\boldsymbol{\sigma} = \frac{2}{J} \mathbf{F} \frac{\partial \hat{W}}{\partial \mathbf{C}} \mathbf{F}^T, \quad (2.1)$$

where  $\boldsymbol{\sigma}$  is the Cauchy stress. In anticipation of formulating a theory of slightly compressible materials, *but without loss of generality*, let the strain energy be alternatively considered as a function of  $J$  and  $\mathbf{C}^* = J^{-2/3} \mathbf{C}$ , i.e.,

$$\hat{W}(\mathbf{C}) = W(J, \mathbf{C}^*), \quad (2.2)$$

assuming that

$$\hat{W}(\mathbf{I}) = W(1, \mathbf{I}) = 0,$$

to ensure zero strain energy in the reference configuration, where here, and in what follows,  $\mathbf{I}$  denotes the appropriate second-order identity tensor. Noting that

$$\frac{\partial J}{\partial \mathbf{C}} = \frac{1}{2} J \mathbf{C}^{-1}, \quad \frac{\partial \mathbf{C}^*}{\partial \mathbf{C}} = -\frac{1}{3} J^{-2/3} \mathbf{C} \otimes \mathbf{C}^{-1} + J^{-2/3} \frac{\partial \mathbf{C}}{\partial \mathbf{C}},$$

the constitutive law (2.1) can therefore be rewritten in the form

$$\begin{aligned} \boldsymbol{\sigma} &= \frac{2}{J} \mathbf{F} \left( \frac{\partial W}{\partial J} \frac{\partial J}{\partial \mathbf{C}} + \frac{\partial W}{\partial \mathbf{C}^*} \frac{\partial \mathbf{C}^*}{\partial \mathbf{C}} \right) \mathbf{F}^T \\ &= \frac{\partial W}{\partial J} \mathbf{I} - \frac{2}{3J} \left( \frac{\partial W}{\partial \mathbf{C}^*} : \mathbf{C}^* \right) \mathbf{I} + \frac{2}{J} \mathbf{F}^* \frac{\partial W}{\partial \mathbf{C}^*} \mathbf{F}^{*T}, \end{aligned} \quad (2.3)$$

where  $:$  denotes the inner product and  $\mathbf{F}^* = J^{-1/3} \mathbf{F}$ . To ensure zero stress in the undeformed state it will be assumed that

$$\frac{\partial W}{\partial J}(1, \mathbf{I}) = 0, \quad \frac{\partial W}{\partial \mathbf{C}^*}(1, \mathbf{I}) = \frac{1}{3} \text{tr} \left[ \frac{\partial W}{\partial \mathbf{C}^*}(1, \mathbf{I}) \right] \mathbf{I}. \quad (2.4)$$

103 Given that  $\text{tr} \left( \mathbf{F}^* \frac{\partial W}{\partial \mathbf{C}^*} \mathbf{F}^{*T} \right) = \frac{\partial W}{\partial J} : \mathbf{C}^*$ , it follows immediately that

$$\text{tr } \boldsymbol{\sigma} = 3 \frac{\partial W}{\partial J}. \quad (2.5)$$

104 It is worthwhile emphasising that this results holds for all elastic materials and for all  
 105 deformations. The Cauchy stress can now be additively decomposed into hydrostatic and  
 106 deviatoric stress components as follows:

$$\boldsymbol{\sigma} = \pi \mathbf{I} + \text{dev } \boldsymbol{\sigma} \quad (2.6)$$

107 where

$$\begin{aligned} \pi(J, \mathbf{C}^*) &= \frac{\text{tr } \boldsymbol{\sigma}}{3} = \frac{\partial W}{\partial J}, \\ \text{dev } \boldsymbol{\sigma}(J, \mathbf{F}^*) &= \boldsymbol{\sigma} - \frac{1}{3} \text{tr } \boldsymbol{\sigma} \mathbf{I} = \frac{2}{J} \left( \mathbf{F}^* \frac{\partial W}{\partial \mathbf{C}^*} \mathbf{F}^{*T} - \frac{1}{3} \left( \frac{\partial W}{\partial \mathbf{C}^*} : \mathbf{C}^* \right) \mathbf{I} \right). \end{aligned} \quad (2.7)$$

### 108 **3 The volumetric-isochoric split and hydrostatic ten-** 109 **sion**

110 Consider now the volumetric-isochoric split

$$W(J, \mathbf{C}^*) = f(J) + \mathcal{W}(\mathbf{C}^*), \quad (3.1)$$

111 an assumption widely made when modelling slightly compressible materials, where the  
 112 separate functionals are assumed infinitely differentiable and the initial conditions

$$f(1) = 0, \quad \mathcal{W}(\mathbf{I}) = 0, \quad (3.2)$$

113 are imposed to ensure zero strain energy in the reference configuration. An immediate  
 114 consequence of this decomposition is that the hydrostatic Cauchy stress is now only a

115 function of the volume change and that the deviatoric Cauchy stress is decomposed into  
 116 separate functions of the volume change and the isochoric deformation gradient tensor  
 117  $\mathbf{F}^*$  as follows:

$$\begin{aligned} \pi(J, \mathbf{C}^*) &= f'(J), \\ \text{dev } \boldsymbol{\sigma}(J, \mathbf{F}^*) &= \frac{2}{J} \boldsymbol{\sigma}_d(\mathbf{F}^*), \quad \boldsymbol{\sigma}_d(\mathbf{F}^*) \equiv \mathbf{F}^* \frac{d\mathcal{W}(\mathbf{C}^*)}{d\mathbf{C}^*} \mathbf{F}^{*T} - \frac{1}{3} \frac{d\mathcal{W}(\mathbf{C}^*)}{d\mathbf{C}^*} : \mathbf{C}^* \mathbf{I}, \end{aligned} \quad (3.3)$$

118 where the prime notation denotes differentiation with respect to the appropriate argument  
 119 and, to ensure that the reference configuration is stress-free, it will be assumed that

$$f'(1) = 0, \quad \text{dev } \boldsymbol{\sigma}_d(\mathbf{I}) = \mathbf{0}. \quad (3.4)$$

120 This decoupling of the hydrostatic stress from the isochoric strain  $\mathbf{C}^*$  and the factorisation  
 121 of the deviatoric stress seem overly prescriptive for anisotropic materials.

Assume now a state of hydrostatic tension, with  $\boldsymbol{\sigma} = \omega \mathbf{I}$ , for which, by definition,

$$\pi(J, \mathbf{C}^*) = \omega, \quad \text{dev } \boldsymbol{\sigma}(J, \mathbf{F}^*) = \mathbf{0}.$$

122 Substitution into (3.3) yields

$$\omega = f'(J), \quad \mathbf{0} = \frac{2}{J} \boldsymbol{\sigma}_d(\mathbf{F}^*). \quad (3.5)$$

123 The first of these determines the volume change in terms of the amount of hydrostatic  
 124 tension. It follows from the initial condition (3.4)<sub>2</sub> that a solution of the second equation  
 125 is given by  $\mathbf{F}^* = \mathbf{I}$ , which is a uniform dilatation. Therefore a material with a separable  
 126 strain energy can behave as if it were an isotropic material under hydrostatic tension, *ir-*  
 127 *respective of the assumed material symmetry*. For infinitesimal strains, uniform dilatation  
 128 is the *unique* solution to the problem of hydrostatic tension for all such elastic materials  
 129 and it seems that the commercial Finite Element codes understandably step this uniform

130 dilation behaviour for infinitesimal strains into the non-linear regime. This explains the  
 131 deficiencies of the assumption (3.1) that have been highlighted elsewhere. The main ar-  
 132 gument advanced by Ní Annaidh *et al.* [4], Vergori *et al.* [6], Gilchrist *et al.* [7] and Nolan  
 133 *et al.* [8], for example, is that Finite Element simulations of the mechanical response to  
 134 hydrostatic tension using specific forms of anisotropy that assume (3.1) yield a purely  
 135 isotropic strain response, with therefore no contribution from, for example, fibres, which  
 136 are the components typically inducing anisotropy in non-linear materials. This seems  
 137 unacceptable physically, since fibres, for example, are much stiffer than the matrix in  
 138 which they are embedded. In light of the analysis presented here, this behaviour is not  
 139 now unexpected. All slightly compressible elastic materials modelled using the additive  
 140 split (1.1) behave isotropically under hydrostatic tension and compression.

141 The mechanical response of bodies for which the hydrostatic stress depends only on  
 142 the volume change can be summarised in the following equivalence theorem, a trivial  
 143 consequence of the identity (2.5), which generalises previous results of, for example,  
 144 Richter [12], Sansour [9] and Horgan and Murphy [10] for specific material symmetries:

**Theorem.** *A strain-energy function has the additive decomposition*

$$W(J, \mathbf{C}^*) = f(J) + \mathcal{W}(\mathbf{C}^*),$$

145 *iff*

$$\text{tr } \boldsymbol{\sigma} = F(J), \quad \text{arbitrary } F(.). \quad (3.6)$$

*Proof.* Assume that (1.1) holds. Then it follows from (2.5) that

$$\text{tr } \boldsymbol{\sigma} = 3f'(J).$$

146 Setting  $F(J) = 3f'(J)$  recovers (3.6).



Now assume that (3.6) holds. The identity (2.5) now yields

$$3\frac{\partial W}{\partial J}(J, \mathbf{C}^*) = F(J).$$

147 A simple integration yields (1.1), with  $f(J) = \int (1/3)F(J)dJ$ . □

148 This equivalence result was first obtained by Charrier *et al.* [5] using a different  
 149 method. It is worthwhile emphasising that this equivalence holds for all deformations of  
 150 all (hyper)elastic materials. The seemingly intuitive appeal of (3.1), and this is surely  
 151 another reason for its widespread adoption, is undermined by this equivalence relation:  
 152 if one wanted to model the slight compressibility of elastic materials, one would surely  
 153 not assume *ab initio* that  $\text{tr } \boldsymbol{\sigma} = F(J)$ , especially when considering anisotropic materials.  
 154 Thus one is lead by consideration of hydrostatic tension to require that either  $\frac{\partial \text{tr } \boldsymbol{\sigma}}{\partial \mathbf{C}^*} \neq \mathbf{0}$ ,  
 155 or, equivalently,

$$\frac{\partial^2 W}{\partial J \partial \mathbf{C}^*} \neq \mathbf{0}, \tag{3.7}$$

156 when modelling the slight compressibility of anisotropic materials.

157 Many of the results obtained here were previously obtained by Federico [2] but the  
 158 approach and emphases here are different.

## 159 4 Generalising the strain energy function

160 There are two approaches that can be adopted to improve the standard model (1.1). One  
 161 could generalise the assumption on the hydrostatic stress (3.6) to include a contribution  
 162 from the isochoric strain tensor or, more immediately, one could generalise the form of the  
 163 strain energy function. Because of the theorem of the last section, these two approaches  
 164 are essentially equivalent and therefore only the second approach will be considered here.  
 165 The method of Spencer [11] is adopted, one that explicitly utilises the fact that the  
 166 volume changes are assumed infinitesimal for slightly compressible materials. Specifically,  
 167 the strain energy function  $W(J, \mathbf{C}^*)$  is expanded in a Taylor series about  $J = 1$  to the

168 second power in  $J - 1$  as follows:

$$W(J, \mathbf{C}^*) = W(1, \mathbf{C}^*) + (J - 1) \frac{\partial W}{\partial J}(1, \mathbf{C}^*) + \frac{1}{2}(J - 1)^2 W_{JJ}(1, \mathbf{C}^*). \quad (4.1)$$

169 Truncation after the second order term seems reasonable, assuming that the  $W_{JJ}$  term  
 170 is of the order of the bulk modulus  $\kappa$ , with all other coefficients of the volumetric terms  
 171 assumed to be of an order less than or equal to this term. Indeed the  $W(1, \mathbf{C}^*)$  and the  
 172  $\frac{\partial W}{\partial J}(1, \mathbf{C}^*)$  terms here are assumed to be of the order of a typical shear modulus of the  
 173 material,  $\mu$ , with the volume change assumed to be of  $\mathcal{O}(\mu/\kappa)$ . Thus, if the strain-energy  
 174 function is non-dimensionalised with respect to the shear modulus, the first term in the  
 175 Taylor series expansion can be considered as the zero-order term and the remaining terms  
 176 the first-order terms in a perturbation series in the parameter  $\mu/\kappa$ .

177 Applying (1.2) to (4.1) yields  $W(1, \mathbf{C}^*) = W_i(\mathbf{C}^*)$ , noting that  $W_i(\mathbf{C})$  is a known in-  
 178 compressible strain-energy function. For convenience, let  $\mathcal{F}(\mathbf{C}^*) \equiv \frac{\partial W}{\partial J}(1, \mathbf{C}^*)$ ,  $\mathcal{G}(\mathbf{C}^*) \equiv$   
 179  $W_{JJ}(1, \mathbf{C}^*)$  and the proposed model therefore has the form

$$W(J, \mathbf{C}^*) = W_i(\mathbf{C}^*) + (J - 1)\mathcal{F}(\mathbf{C}^*) + \frac{1}{2}(J - 1)^2\mathcal{G}(\mathbf{C}^*). \quad (4.2)$$

Some rationale for the choice of  $\mathcal{F}, \mathcal{G}$  is needed if (4.2) is to be a workable model of slight  
 compressibility. Some preliminary guidance is given by the restrictions that should be  
 imposed on every candidate strain energy. First note that if  $W_i(\mathbf{I}) = 0$ , then the strain  
 energy is zero in the reference configuration. Substituting (4.2) into the initial conditions  
 (2.4) for zero stress in the reference configuration yields

$$\mathcal{F}(\mathbf{I}) = 0, \quad W'_i(\mathbf{I}) = \frac{1}{3} \text{tr} [W'_i(\mathbf{I})] \mathbf{I}.$$

The compressibility condition (3.7) requires that

$$\mathcal{F}'(\mathbf{I}) \neq 0.$$

180 Therefore  $\mathcal{F}(\mathbf{C}^*)$  cannot be constant and, in particular, cannot be identically zero, as  
 181 is assumed in the separable model of slight compressibility, (3.1). Motivated by a desire  
 182 to keep mathematical models as simple as possible, it is therefore proposed that  $\mathcal{F}(\mathbf{C}^*)$   
 183 be a *linear* function of the appropriate invariants of  $\mathbf{C}^*$ . Comparing the first two terms  
 184 in (4.2) shows that the complexity of the model in  $\mathbf{C}^*$  is reduced when we include the  
 185 linear term in the volume change. Keeping this structure in mind, it will be additionally  
 186 assumed that  $\mathcal{G}(\mathbf{C}^*)$  is a positive constant ( $= c$ ) and therefore the proposed model of  
 187 slight compressibility has the form

$$W(J, \mathbf{C}^*) = W_i(\mathbf{C}^*) + (J - 1)\mathcal{F}(\mathbf{C}^*) + \frac{c}{2}(J - 1)^2, \quad (4.3)$$

with, from (2.5),

$$\text{tr } \boldsymbol{\sigma} = \mathcal{F}(\mathbf{C}^*) + c(J - 1).$$

188 Some additional simplicity in the linear form of  $\mathcal{F}(\mathbf{C}^*)$  is assumed here. Recalling  
 189 the centrality of the problem of hydrostatic tension in the argument for a new account-  
 190 ing for slight compressibility, it is proposed that  $\mathcal{F}(\mathbf{C}^*)$  is linear in the smallest subset  
 191 of invariants in  $\mathbf{C}^*$  that ensure that the hydrostatic stress for slight compressibility is  
 192 compatible with the hydrostatic stress for the linear theory, on restriction to infinites-  
 193 imal strains. Examples of the application of this procedure will be given in the next  
 194 section for isotropic, transversely isotropic and orthotropic materials. The reasons for  
 195 this are twofold: (1) every well-posed non-linear theory should recover its linear form on  
 196 restriction to infinitesimal inputs (Quintanilla and Saccomandi [3]) and (2) if a slightly  
 197 compressible material is locally subjected to hydrostatic tension in simulations of practi-  
 198 cal problems, for which the applied stresses will be of the order of a typical shear modulus  
 199 of the material, the linear theory becomes applicable as the volume change is assumed  
 200 infinitesimal.

201 To determine the material constants in (4.3), it is proposed that experiments where  
 202 the principal Cauchy stresses are known, such as uniaxial and biaxial tension, be per-

203 formed with the volume change, the strains and all other geometrical and kinematical  
 204 variables necessary to determine the appropriate functions of  $\mathbf{C}^*$  measured *simultane-*  
 205 *ously*. Simple regression analysis then determines the material constants. To the best of  
 206 the authors' knowledge such a comprehensive, simultaneous collection of data has never  
 207 been performed for anisotropic, soft tissue but is essential if reliable models of its mechan-  
 208 ical response are to be obtained. Some of these variables have been measured in isolation  
 209 from the others. For example, there is a vast literature on the biaxial testing of soft tissue  
 210 for which the strains alone have been measured. An indication of the state of the art is the  
 211 collection of data of Holzapfel [15], who, in addition to measuring the orientation of the  
 212 collagen fibres inducing anisotropy in arterial tissue, also obtained uniaxial stress-strain  
 213 data in both the axial and circumferential directions for the human aorta. The essential  
 214 missing component is, of course, measurement of the volume change. The analysis here  
 215 suggests that simultaneous measurement of the volume change together with all the other  
 216 necessary variables is an essential first step in the development of accurate and reliable  
 217 models of slight compressibility.

## 218 5 Some examples of material symmetries

To illustrate the ideas of the last section for the new model of slight compressibility pro-  
 posed here, appropriate forms of slight compressibility for isotropic, transversely isotropic  
 and orthotropic materials will now be considered. A key feature of the proposed model is  
 that the linear theory in each case is recovered on restriction to infinitesimal deformations.  
 To consider *infinitesimal* strains, assume that

$$\mathbf{F} = \mathbf{I} + \mathbf{H}, \quad h \equiv \sqrt{\mathbf{H} : \mathbf{H}} \ll 1,$$

where  $\mathbf{H}$  is the displacement gradient tensor. Then, neglecting here and hereafter higher  
 order terms,

$$J = 1 + \text{tr } \boldsymbol{\epsilon},$$

219 where  $\boldsymbol{\epsilon} = 1/2 (\mathbf{H} + \mathbf{H}^T)$  is the infinitesimal strain tensor. It follows that, to the first-  
 220 order term in  $\mathbf{H}$ ,

$$\begin{aligned} \mathbf{F}^* &= J^{-1/3} \mathbf{F} = \mathbf{I} + \mathbf{H}^*, & \mathbf{H}^* &\equiv \mathbf{H} - \frac{1}{3} \text{tr } \boldsymbol{\epsilon} \mathbf{I}, \\ \mathbf{C}^* &= J^{-2/3} \mathbf{C} = \mathbf{I} + 2\boldsymbol{\epsilon}^*, & \boldsymbol{\epsilon}^* &\equiv \boldsymbol{\epsilon} - \frac{1}{3} \text{tr } \boldsymbol{\epsilon} \mathbf{I}. \end{aligned} \quad (5.1)$$

## 221 5.1 Isotropic materials

222 The strain energy function for isotropic materials can be written in the form  $W =$   
 223  $W(J, I_1^*, I_2^*)$ , where

$$I_1^* = \text{tr } \mathbf{C}^*, \quad I_2^* = \frac{1}{2} ((\text{tr } \mathbf{C}^*)^2 - \text{tr } (\mathbf{C}^*)^2). \quad (5.2)$$

224 Since  $\mathcal{F}(\mathbf{C}^*)$  is assumed linear in the invariants of  $\mathbf{C}^*$  the hydrostatic stress becomes

$$\text{tr } \boldsymbol{\sigma} = c_1(I_1^* - 3) + c_2(I_2^* - 3) + c(J - 1). \quad (5.3)$$

On restriction to infinitesimal deformations and truncating after first order terms in  $h$ ,

$$\text{tr } \boldsymbol{\sigma} = c \text{tr } \boldsymbol{\epsilon}.$$

The linear theory for isotropic material yields

$$\text{tr } \boldsymbol{\sigma} = (3\lambda + 2\mu) \text{tr } \boldsymbol{\epsilon},$$

where  $\lambda, \mu$  are the Lamé constants. A comparison of these two linear forms for the hydrostatic stress shows that the simplest form of the non-linear hydrostatic stress (5.3) that is compatible with the linear is the choice

$$c_1 = c_2 = 0, \quad c = 3\lambda + 2\mu,$$

225 and therefore for isotropic materials the proposed slightly compressible strain energy that  
 226 is of the prototypical, general form (4.2) is given by setting  $\mathcal{F}(\mathbf{C}^*) \equiv 0$  and  $\mathcal{G}(\mathbf{C}^*) =$   
 227  $c = 3\lambda + 2\mu$ , i.e.,

$$W(J, \mathbf{C}^*) = W_i(\mathbf{C}^*) + \frac{3\lambda + 2\mu}{2}(J - 1)^2, \quad (5.4)$$

228 which has the separable form (1.1) and therefore the volumetric-isochoric split seems  
 229 acceptable for isotropic materials.

## 230 5.2 Transversely isotropic materials

231 For these materials the general strain energy has the form  $W = W(J, I_1^*, I_2^*, I_4^*, I_5^*)$ , where  
 232  $I_1^*, I_2^*$  are given in (5.2) and

$$I_4^* = \mathbf{M} \cdot \mathbf{C}^* \mathbf{M}, \quad I_5^* = \mathbf{M} \cdot (\mathbf{C}^*)^2 \mathbf{M}, \quad (5.5)$$

233 where  $\mathbf{M}$  is the preferred direction of the materials. The initial assumption for  $\mathcal{F}(\mathbf{C}^*)$   
 234 is therefore that

$$\mathcal{F}(\mathbf{C}^*) = c_1(I_4^* - 1) + c_2(I_5^* - 1), \quad (5.6)$$

235 noting that there are no terms linear in the isotropic invariants following the analysis  
 236 of the last subsection and therefore now

$$\text{tr } \boldsymbol{\sigma} = c_1(I_4^* - 1) + c_2(I_5^* - 1) + c(J - 1). \quad (5.7)$$

On restriction to infinitesimal deformations, the hydrostatic stress is therefore

$$\text{tr } \boldsymbol{\sigma} = \left( c - \frac{2}{3}c_1 - \frac{4}{3}c_2 \right) \text{tr } \boldsymbol{\epsilon} + 2(c_1 + 2c_2) \mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M}$$

The linear theory for transversely isotropic materials (see, for example, Spencer [16]) that

$$\text{tr } \boldsymbol{\sigma} = (3\lambda + 2\mu_T + \alpha) \text{tr } \boldsymbol{\epsilon} + (3\alpha + \beta + 4[\mu_L - \mu_T]) \mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M},$$

237 using Spencer's notation for the five material constants. A comparison of the two forms  
 238 of the hydrostatic stress shows that one of the constants  $c_1, c_2$  in the provisional linear  
 239 expansion (5.6) can be set equal to zero. Thus there are two slightly compressible forms  
 240 consistent with the linear theory, i.e.,

$$\begin{aligned} W(J, \mathbf{C}^*) &= W_i(\mathbf{C}^*) + c_1(J-1)(I_4^* - 1) + \frac{c}{2}(J-1)^2, \\ W(J, \mathbf{C}^*) &= W_i(\mathbf{C}^*) + c_1(J-1)(I_5^* - 1) + \frac{c}{2}(J-1)^2, \end{aligned} \quad (5.8)$$

241 where the constants  $c_1, c$  are to be determined from volume change measurements during  
 242 characterisation tests.

### 243 **5.3 Materials with two families of mechanically equivalent fi-** 244 **bres**

Denote the two preferred directions by  $\mathbf{M}, \mathbf{M}'$ , with now  $W = W(J, I_1^*, I_2^*, I_4^*, I_5^*, I_6^*, I_7^*, I_8^*)$ ,  
 where  $I_1^*, I_2^*, I_4^*, I_5^*$  have been defined previously and

$$I_6^* = \mathbf{M}' \cdot \mathbf{C}^* \mathbf{M}', \quad I_7^* = \mathbf{M}' \cdot (\mathbf{C}^*)^2 \mathbf{M}', \quad I_8^* = \mathbf{M} \cdot \mathbf{M}' \mathbf{M} \cdot \mathbf{C}^* \mathbf{M}'.$$

245 Bearing in mind that the fibres are mechanically equivalent, assume initially therefore  
 246 that

$$\mathcal{F}(\mathbf{C}^*) = c_1(I_4^* + I_6^* - 2) + c_2(I_5^* + I_7^* - 2) + c_3(I_8^* - (\mathbf{M} \cdot \mathbf{M}')^2), \quad (5.9)$$

247 with now

$$\text{tr } \boldsymbol{\sigma} = c_1(I_4^* + I_6^* - 2) + c_2(I_5^* + I_7^* - 2) + c_3(I_8^* - (\mathbf{M} \cdot \mathbf{M}')^2) + c(J-1). \quad (5.10)$$

248 On restriction to infinitesimal deformations therefore

$$\begin{aligned}
\text{tr } \boldsymbol{\sigma} &= 2(c_1 + 2c_2)(\mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M} + \mathbf{M}' \cdot \boldsymbol{\epsilon} \mathbf{M}') - \frac{2}{3} \text{tr } \boldsymbol{\epsilon} + 2c_3 \mathbf{M} \cdot \mathbf{M}' (\mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M}' - \frac{1}{3} \mathbf{M} \cdot \mathbf{M}' \text{tr } \boldsymbol{\epsilon}) + c \text{tr } \boldsymbol{\epsilon} \\
&= \text{tr } \boldsymbol{\epsilon} \left( c - \frac{4}{3} (c_1 + 2c_2) - \frac{2}{3} c_3 (\mathbf{M} \cdot \mathbf{M}')^2 \right) + 2(c_1 + 2c_2) (\mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M} + \mathbf{M}' \cdot \boldsymbol{\epsilon} \mathbf{M}') \\
&\qquad\qquad\qquad + 2c_3 \mathbf{M} \cdot \mathbf{M}' \mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M}'. \tag{5.11}
\end{aligned}$$

249 In the corresponding linear theory (see, for example, Equation (45) of Spencer [16] and  
250 using his notation)

$$\begin{aligned}
\text{tr } \boldsymbol{\sigma} &= (3\lambda + 2\mu_T + 2\gamma_3 + \gamma_4 (\mathbf{M} \cdot \mathbf{M}')^2) \text{tr } \boldsymbol{\epsilon} + \\
&(\mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M} + \mathbf{M}' \cdot \boldsymbol{\epsilon} \mathbf{M}') (2\gamma_1 + 3\gamma_3 + \gamma_5 (\mathbf{M} \cdot \mathbf{M}')^2 + \gamma_6 + 2\gamma_7) + \\
&(\gamma_2 \mathbf{M} \cdot \mathbf{M}' + 2\gamma_5 + 3\gamma_6) \mathbf{M} \cdot \mathbf{M}' \mathbf{M} \cdot \boldsymbol{\epsilon} \mathbf{M}'. \tag{5.12}
\end{aligned}$$

Comparing (5.11)<sub>2</sub> and (5.12) shows that to fully recover the linear theory, it is required that

$$c_3 \neq 0,$$

251 and that one of  $c_1, c_2$  can be set equal to zero, in order to simplify the complexity of the  
252 model. Thus, adopting an obvious change of notation for the material constants, either  
253 of

$$\begin{aligned}
W(J, \mathbf{C}^*) &= W_i(\mathbf{C}^*) + c_1(J-1)(I_4^* + I_6^* - 2) + c_3(J-1) \left( I_8^* - (\mathbf{M} \cdot \mathbf{M}')^2 \right) + \frac{c}{2}(J-1)^2, \\
W(J, \mathbf{C}^*) &= W_i(\mathbf{C}^*) + c_1(J-1)(I_5^* + I_7^* - 2) + c_3(J-1) \left( I_8^* - (\mathbf{M} \cdot \mathbf{M}')^2 \right) + \frac{c}{2}(J-1)^2, \tag{5.13}
\end{aligned}$$

254 seems a reasonable model of slight compressibility for orthotropic materials, adopting the  
255 motivation proposed earlier. Note the necessity of including an  $I_8^*$  term in the proposed  
256 model for slight compressibility, a term which is absent in the model of Nolan *et al.* [8],  
257 for example.



## 258 Acknowledgements

259 The authors would like to thank Professor P. Neff for his insight and encouragement. The  
260 authors are also grateful to the anonymous referees whose comments have resulted in a  
261 significantly improved final version of this paper.

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